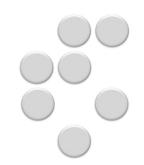
# HOW CAN PLASMAS BUILD CARBON NANOSTRUCTURES?



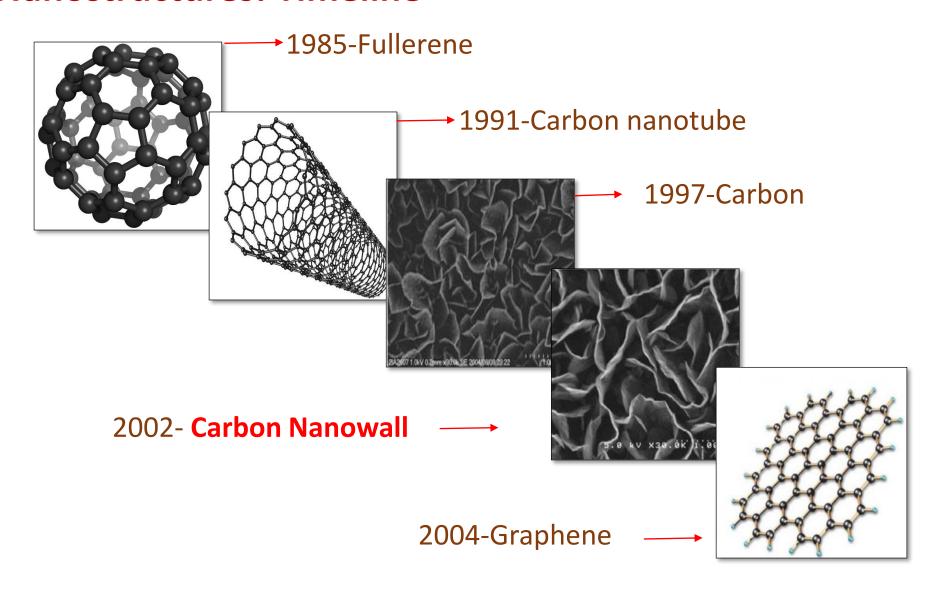
**Uroš Cvelbar**, Neelakandan M. Santhosh, Gregor Filipič, Oleg Baranov *Jozef Stefan Institute, Ljubljana, Slovenia.* 



M. Hori, M. Sekine, H. Kondo

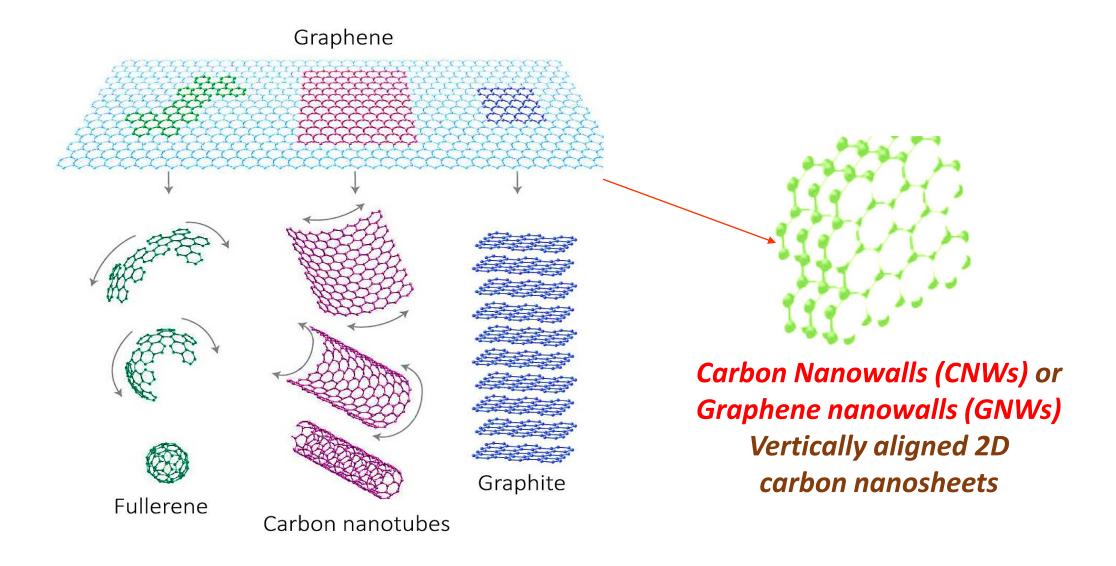


#### **Carbon Nanostructures: Timeline**

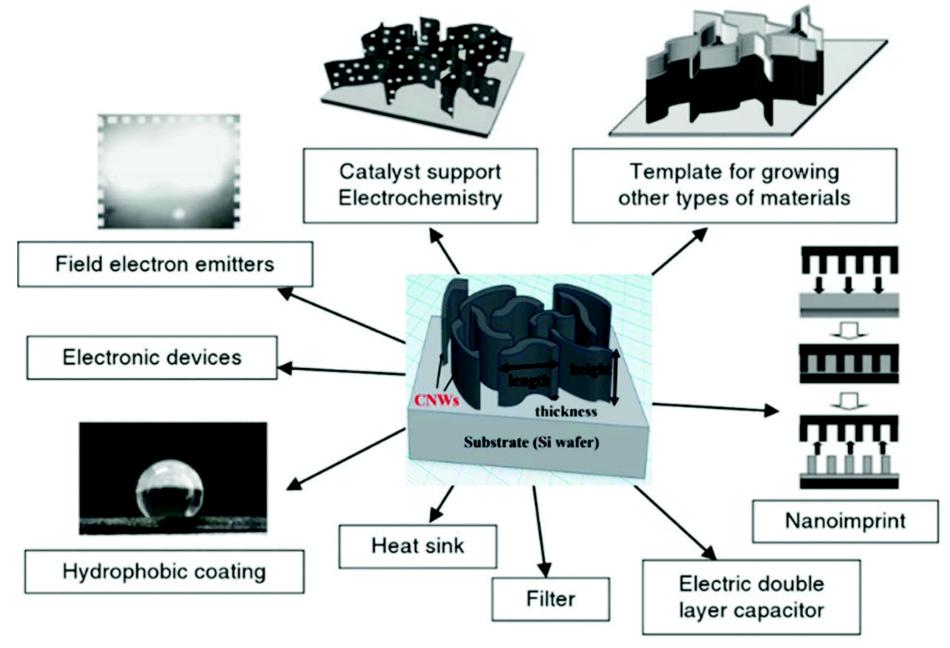


- 1. Ando, Y.; Zhao, X.; Ohkohchi, M. *Carbon N. Y.* **1997**, *35*, 153–158, doi:10.1016/S0008-6223(96)00139-X.
- 2. Wu, Y.; Qiao, P.; Chong, T.; Shen, Z. Adv. Mater. **2002**, *14*, 64–67, doi:10.1002/1521-4095(20020104)14:1<64::AID-ADMA64>3.0.CO;2-G.

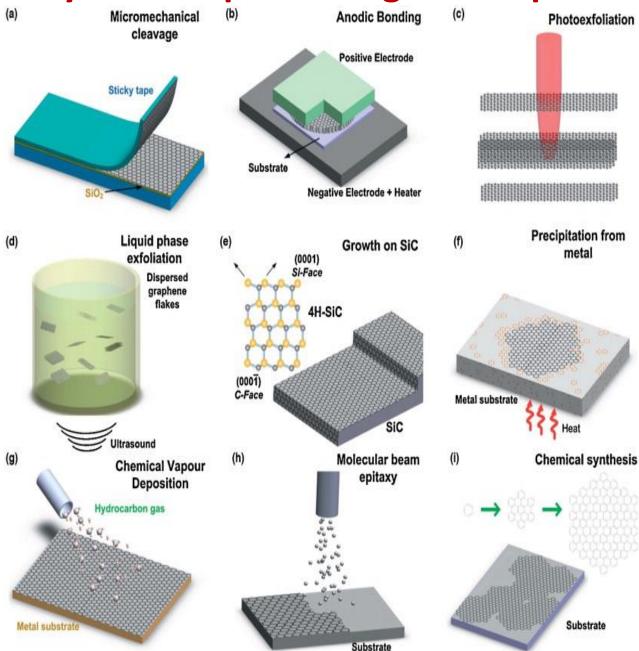
#### **Graphene: Constituent of Carbon Nanostructures**



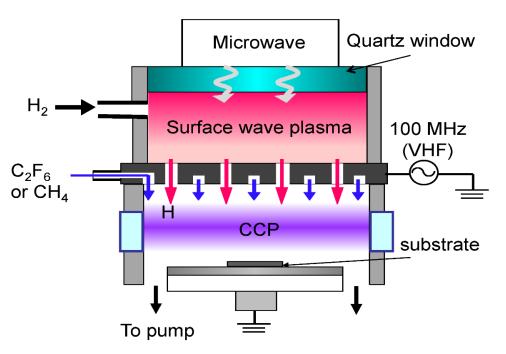
#### Why Carbon Nanowalls...?



#### Why Plasma-processing Techniques..?



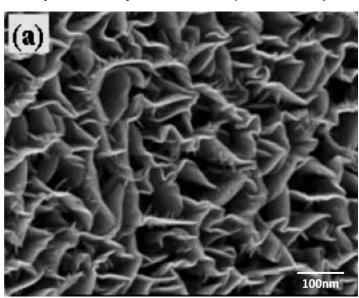
#### Plasma assisted method



- Large scale growth with selected properties
- **Control** in ordering/patterning
- Rapid production and fast processing
- Easier and safer
- 1. Y. L. Zhong, Z. Tian, G. P. Simon, and D. Li, Scalable production of graphene via wet chemistry: Progress and challenges, (2015).
- 2. O. Gökhan Eğilmez, Gürsel A. Süer,Özgüner, *Des. Control Appl. Mechatron. Syst. Eng.* 135 (2012).doi:10.5772/67458

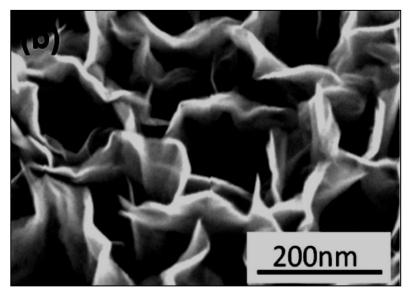
#### **Plasma-assisted Synthesis Methods**

- > Bottom-up approch
- Hydrocarbon/ Fluorocarbon gas sources
- Plasma interaction, depositionnucleation and growth
- Plasma Enhanced Chemical Vapor Deposition (PECVD)



Wu, Y.; Yang, B.; Zong, B.; Sun, H.; Shen, Z.; Feng, Y. J. Mater. Chem. **2004**, 14

- > Top-down approach
- Solid/ Liquid carbon precursor
- Plasma treatment on the precursor and growth
- Plasma treatment on honey



Seo, D. H.; Rider, A. E.; Kumar, S.; Randeniya, L. K.; Ostrikov, K. Carbon N. Y. **2013**, 60, 221–228

#### Plasma Treatment On Solid/Liquid Carbon Sources

Carbon containing
hydrogels- resorcinolformaldehyde, cellulose
hydrogel, phenolformaldehyde

Ar, H<sub>2</sub> gas

Plasma treatment

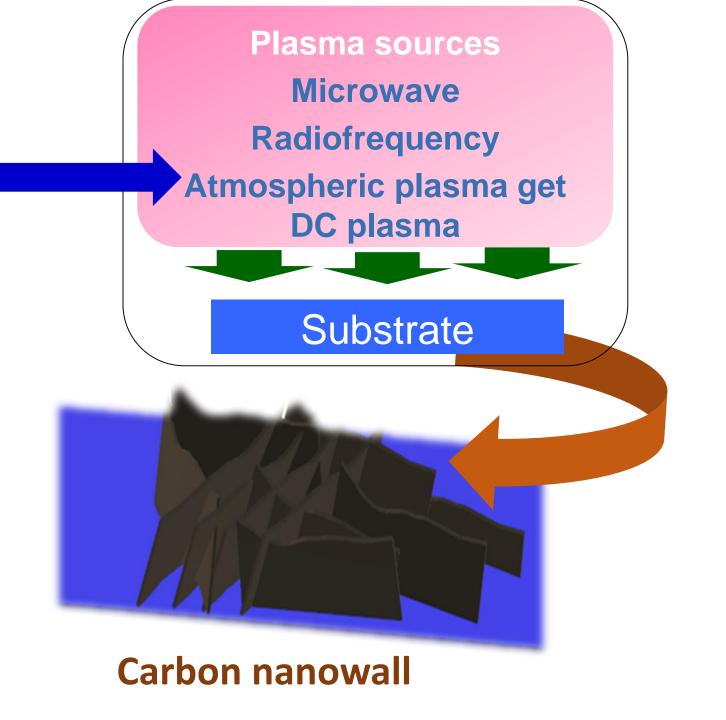
Elevating temperature,
dissociation into
different radicals,
etching of amorphous
phase

Carbon nanowall/
nanostructures with
uniform morphology
and inter layer spacing

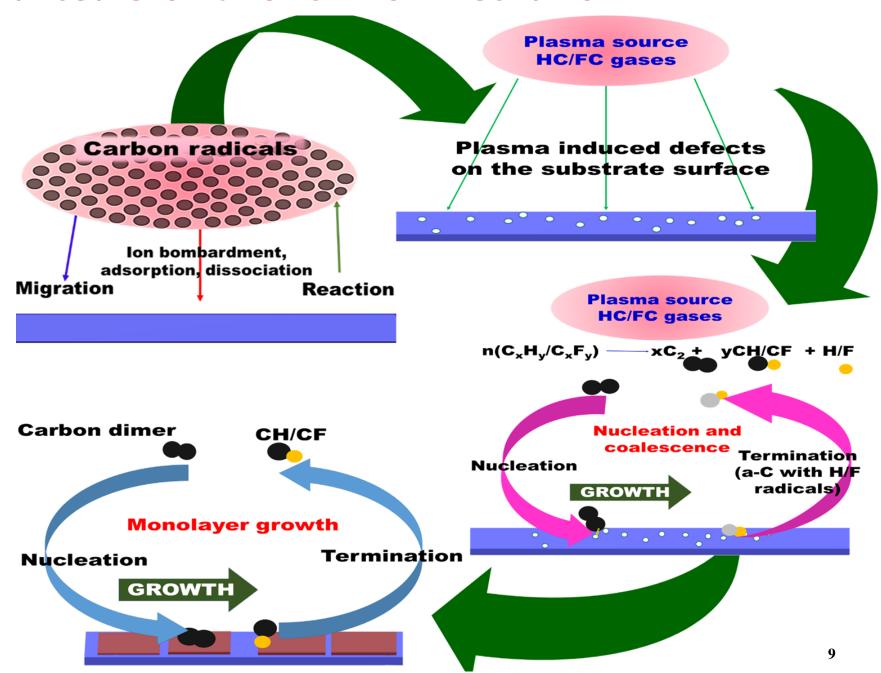
#### **PECVD**

Gas species
CH<sub>4</sub>,C<sub>2</sub>H<sub>2</sub>,
C<sub>2</sub>F<sub>6</sub>,H<sub>2</sub>,Ar,NH<sub>3</sub>

- Vertical growth of CNWs
   (Currently reported maximum ~ 300nm)
- Atomically thin edges, controlled spacing, and excellent height uniformity



#### Plasma Enhanced Growth of CNWs - Mechanism

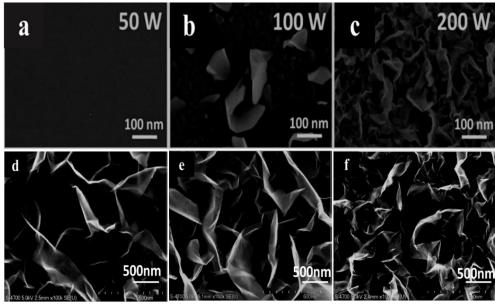


#### **Discharge Parameters**

#### Finding optimum discharge parameters for the high yield growth

#### Different plasma power

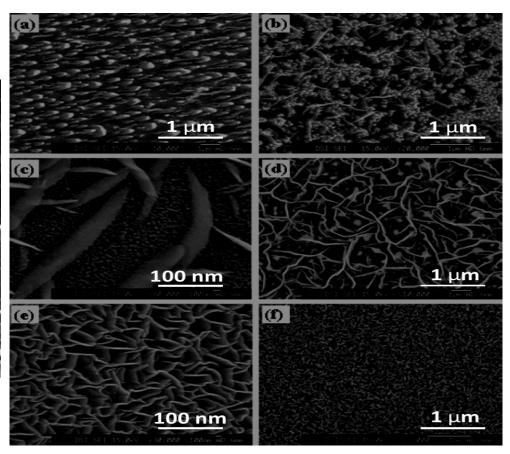
Yang, C.; Bi, H.; Wan, D.; Huang, F.; Xie, X.; Jiang, M. Direct J. Mater. Chem. A **2013**, 1, 770–775,



**CH**<sub>4</sub> concentration

d) 10%, e) 40%, f) 100%

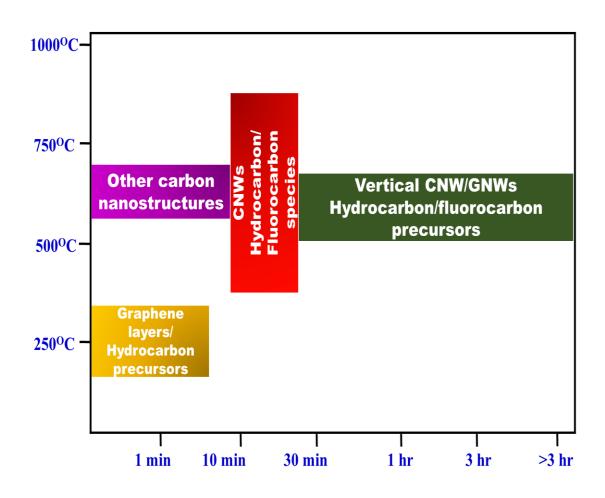
Wang, J.; Zhu, M.; Outlaw, R. A.; Zhao, X.; Manos, D. M.; Holloway, B. C., Carbon N. Y. **2004**, 42, 2867–2872,.



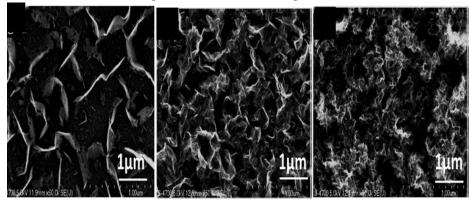
Different H<sub>2</sub>/CH<sub>4</sub> flow rate ratios: (a) 30, (b) 15, (c) 10, (d) 6, (e) 4, (f) 1sccm

Wu, Y.; Yang, B.; Zong, B.; Sun, H.; Shen, Z.; Feng, Y. J. Mater. Chem. **2004**, 14, 469,

#### **Substrate Temperature and Growth Time**



Different growth temperatures: 630 °C, 730 °C, 830 °C.



Wang, J.; Zhu, M.; Outlaw, R. A.; Zhao, X.; Manos, D. M.; Holloway, B. C., Carbon N. Y. **2004**, 42, 2867–2872,.

Attaining the growth of CNWs @ low-temperature and short-time period with high quality

Enhance the thermal stability of CNWs

#### **Effect of Pressure**

#### Lower plasma pressure

- ✓ Higher electron energy
- ✓ Increasing ionization rate
- ✓ A low growth rate
- ✓ Higher vertical orientation

#### Higher plasma pressure

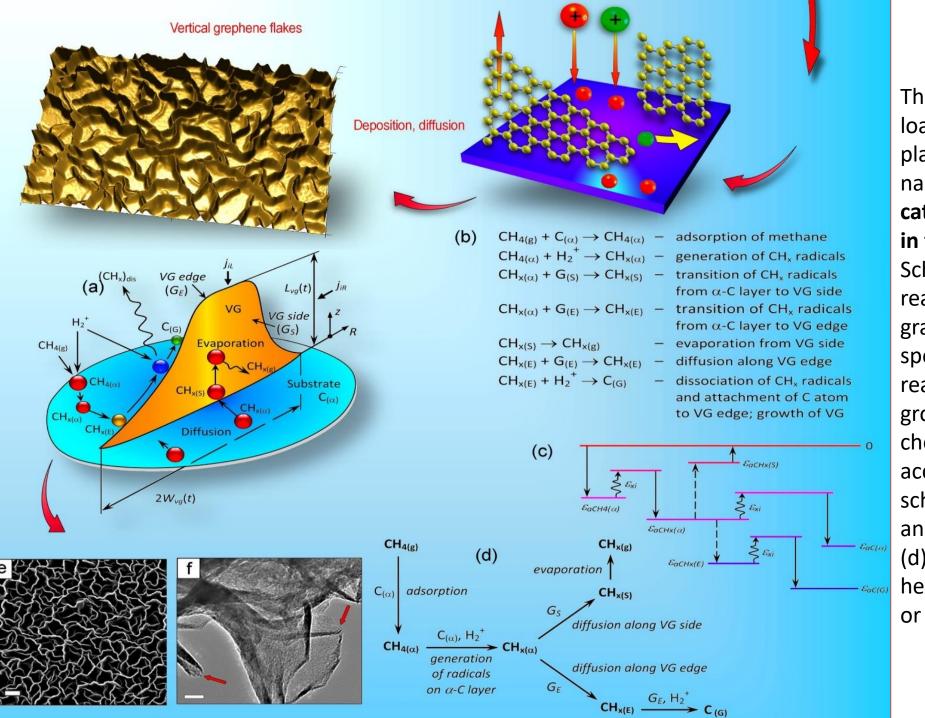
- ✓ Larger volume of feedstock gas input
- ✓ Electron energy decreased
  - ✓ The massive production

Control the pressure, synthesize CNWs with uniform morphology, high orientation and larger interlayer spacing

## Simulations - Formation of Vertically Oriented Graphenes: What are the Drivers of Growth?

We build a multi-scale, multi-factor model which was thoroughly verified by comparison with a large massive of experimental data to ensure a significant chemical and physical insight into the processes that determine nucleation, growth and structure formation of vertically-aligned graphenes from the case of plasma.

The leading role of surface diffusion fluxes, rather than direct influx from gas phase, was confirmed with the ion bombardment being a key factor 'switching' the growth modes by generating surface defects and hence, increasing the surface adsorption energy. Thus, the hydrocarbon radicals generated on a substrate due to the bombardment diffuse to the nanoflakes and catalyze the reactions, and serve as the primary source of material to build the nanoflakes.



The prepared substrates are loaded into the chamber of plasma reactor where the nanoflakes are nucleated on catalyst particles and then grow in the ion and atom flux.

Schematic of the mechanisms and reactions involved in vertical graphene growth. Motion of species involved into the reactions about the surface of growing graphene flake (a); list of chemical reactions taken into account in the model (b); schematic of the reactions (c); and diagram of the energy levels (d). Substrate may be externally heated by the use of heating coil, or heated directly by the plasma.

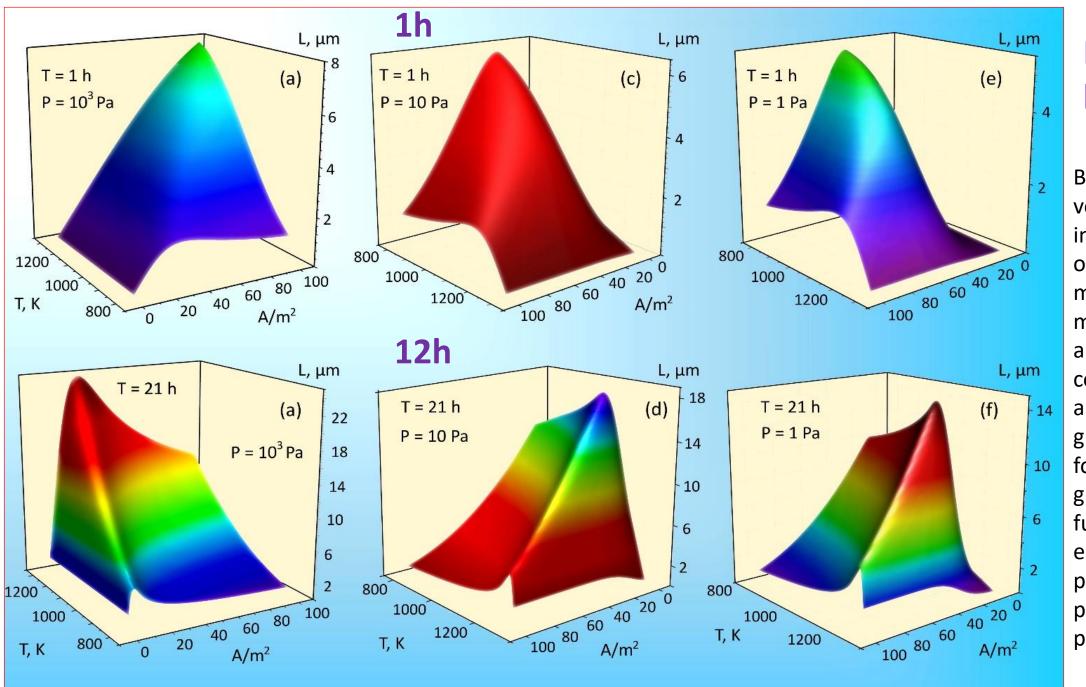
The ion bombardment changes the adsorption mechanism on the substrate where the methane molecules (primary source of CHx radicals) adsorb, from the physical sorption ( $\sim$  0.5 eV) to the chemical sorption ( $\sim$  1-2 eV), to generate the large amount of the radicals on the surface at the elevated temperatures. The radical can diffuse along the surface layer, and can be adsorbed on the GNF flat side or sharp edge during the diffusion, at that the adsorption energy is changed from  $\epsilon_a CH_\chi(\alpha)$  to  $\epsilon_a CH_\chi(S)$  or  $\epsilon_a CH_\chi(E)$ , respectively. If the radical is adsorbed on the high-defect GNF edge, the adsorption energy is increased:

$$(\varepsilon_a CH_X(\alpha) \rightarrow \varepsilon_a CH_X(E)),$$

and the radical can diffuse along the edge with low probability of evaporation or transition to the side surface of the GNF sheet  $(\varepsilon_a CH_X(G) >> \varepsilon_a CH_X(S))$ , see Fig. 1.

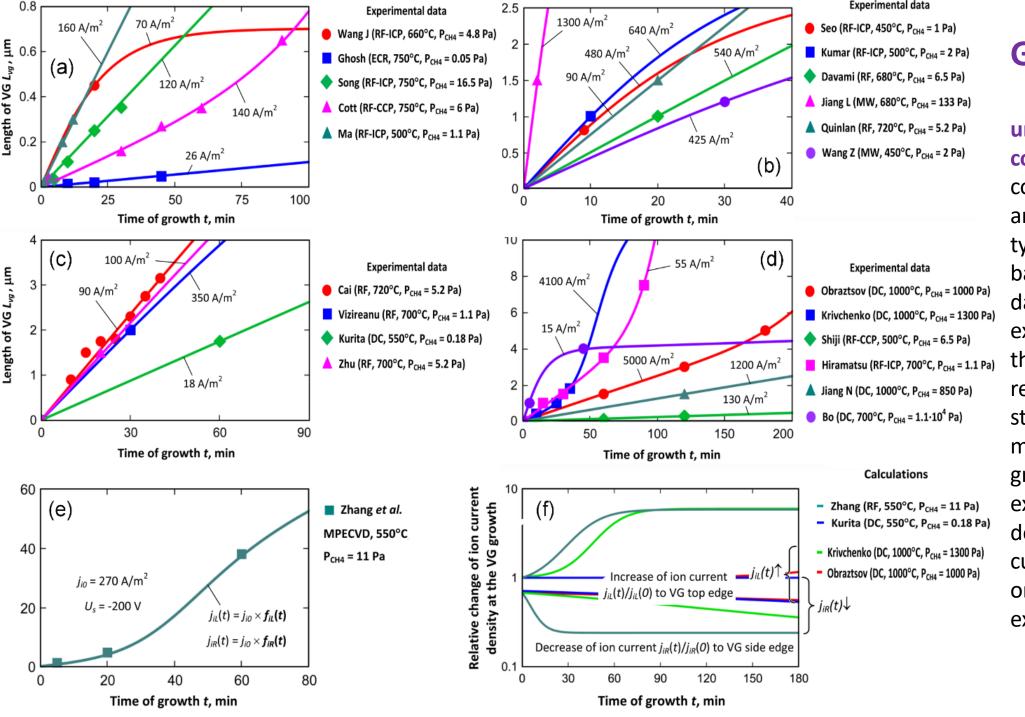
Furthermore, the model implicates several features:

- (i) The nanoflakes morphology was explained by the difference between the energy of the physical sorption  $\varepsilon_a CH_\chi(S)$  on the defect-free flat surface of nanoflake, and the energy  $\varepsilon_a CH_\chi(E)$  of the chemical sorption on the defected edges of the GNF. The GNF grow when  $CH_\chi$  radical is struck by ion at the diffusion from the GNF bottom to the top, and the carbon atom is released.
- (ii) (As suggested by a number of researchers, 14,29 the electric field on the exposed sharp edges of the nanoflakes is taken into account when describing the GNF growth. Strong enhancement of the electric field by sharp edges of a nanoflake plays a vital role in the GNF growth, since it intensifies the reactions by order of a magnitude at the ion focusing.
- (iii) The **diffusion of radicals from the substrate to the GNF top** and the growth of the GNF in width depend on the ion current density  $j_{iR}$  to the side, while the nanoflake growth in height depends on the ion current density  $j_{iR}$  to the top; a re-distribution of the densities occurs at the nanoflakes growth.
- (iv) **The current densities to the side and top edges** of the nanoflake change during the growth because of shielding of the nanoflakes side edge by the neighboring nanoflakes after the GNF array grows high.



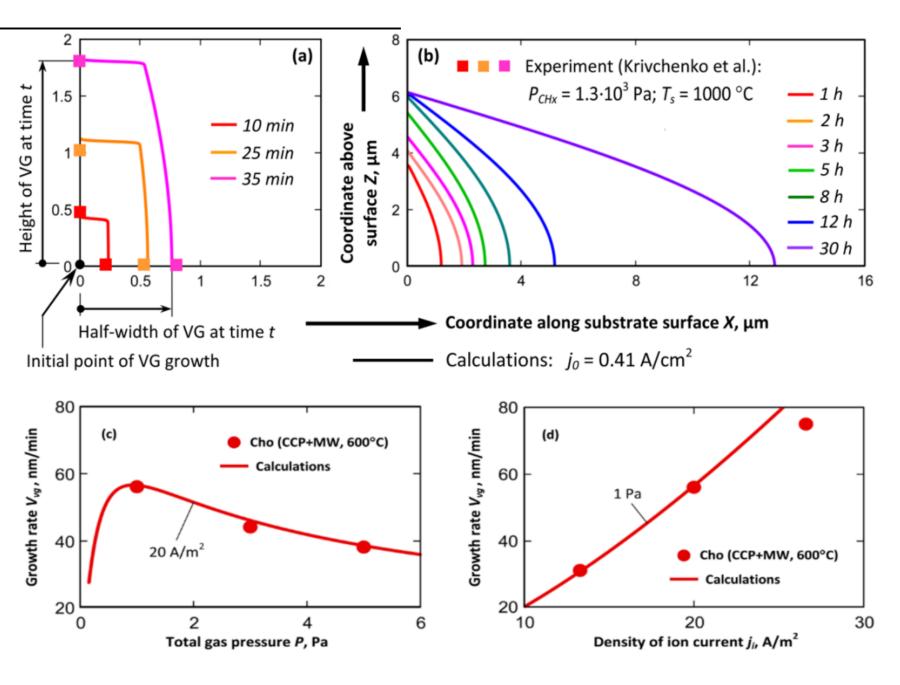
# Nanoflake length

Big data", i.e. large volume of information obtained from the multiscale multifactor model allows comprehensive analyzation of the graphene formation and growth as a function of all essential parameters and processes in plasma.



#### **Growth kinetics**

**GNF** length time on under various growth conditions. was considered a parameter, and calculated for various types of plasma discharge based on the experimental Comparison with experimental data shows that the ion bombardment results in changing the ionstimulating diffusion to ion mixing mode. Non-linear growth with saturation is explained by dependence of the current densities  $j_{iL}$  and  $j_{iR}$ on time for the particular experiment



#### Time evolution of a GNF shape

at  $j_0 = 0.41 \text{ A/m}^2$ , PCHx = 1300 Pa, Ts = 1000 °C. The GNF shape evolves from a rectangular with a height to width ratio of about unity at the initial stages of growth (a), to a ribbon-like structure of limited height  $(\sim 6 \mu m)$  (b). Experimental bars correspond to the results obtained by Krivchenko et al. 18 In the calculations the non-liner fit of jiLjiR is assumed. (c) GNF growth rate on density of ion current, total gas pressure is a parameter.

On the experimental side ...

#### **CH-CNWs** Experimental

#### **CF-CNWs** Experimental

### RI-PECVD system (Radical Injection Plasma Enhanced CVD)

✓  $CH_4$ : 100 sccm  $H_2$ : 50 sccm

✓ Substrate : SiO<sub>2</sub>

✓ Working pressure : 1 Pa

✓ Temperature : 600°C

✓ SWP power: 400 W

✓ CCP power : 300 W

#### Gas flow rates

 $H_2: 100 \text{ sccm } C_2F_6: 50 \text{ sccm}$ 

Pressure: 133 Pa

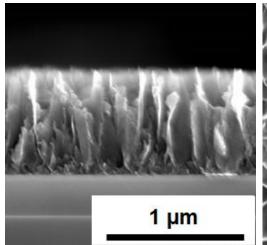
<u>Substrate temperature</u>: 630°C

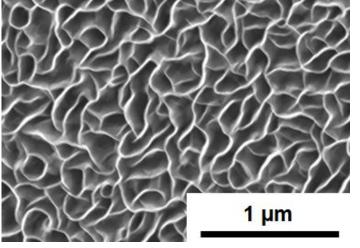
Microwave power: 250 W

VHF power: 250 W

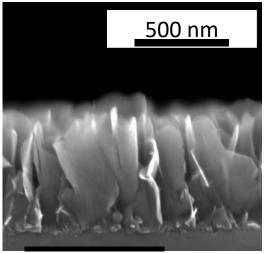
Growth time: 40 min

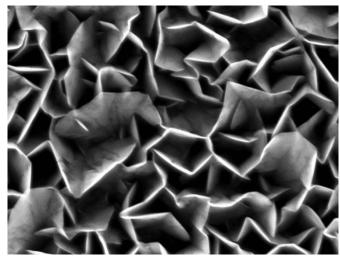
#### **CH-CNWs** SEM images



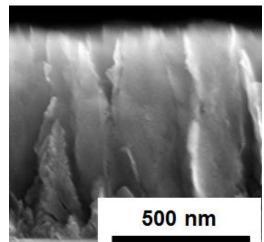


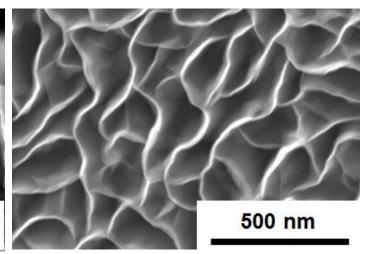
#### CF-CNWs 1

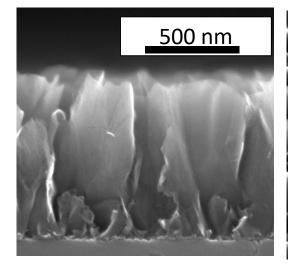


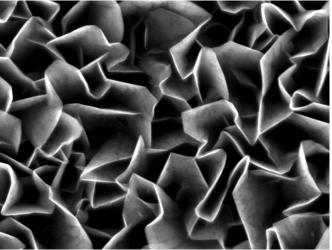


CF-CNWs 2





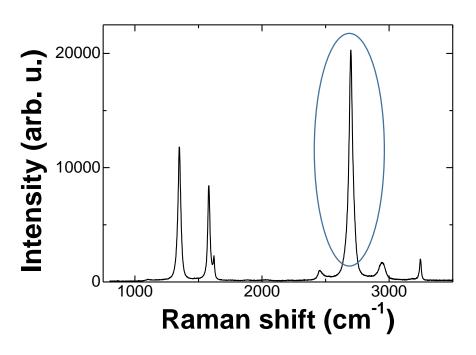




#### **CH-CNWs** Raman spectrum

# Tok 60k 50k Adok 20k 10k 10k 10k 10k Raman shift (cm<sup>-1</sup>)

#### CF-CNWs 1



D band - 1328 cm<sup>-1</sup>

G band - 1584 cm<sup>-1</sup>

D' band - 1620 cm<sup>-1</sup>

2D band - 2685 cm<sup>-1</sup>

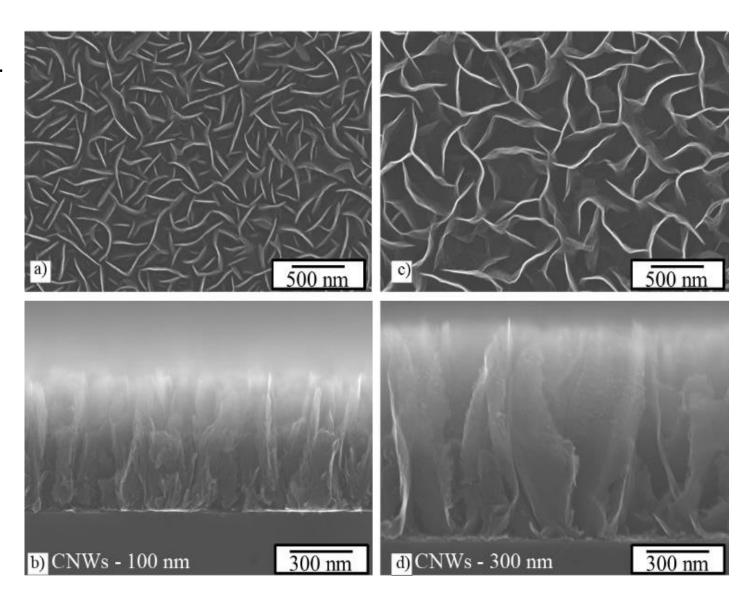
D+G band - 2935 cm<sup>-1</sup>

2D' band – 3230 cm<sup>-1</sup>

#### Different spacing between CNW

Process conditions and results of analyses of both principal tested CNWs (100 nm) and CNWs (300 nm).

Average gap between each CNWs [nm]	100	300
Average height of CNWs [nm]	650	970
Conductivity [S/cm]	38	76
Process pressure [Pa]	1	5
Substrate temperature [°C]	600	600
SWP power [W]	400	400
CCP power [W]	100	500
H <sub>2</sub> [sccm]:CH <sub>4</sub> [sccm]	50:100	50:100
Growth time [min]	60	10

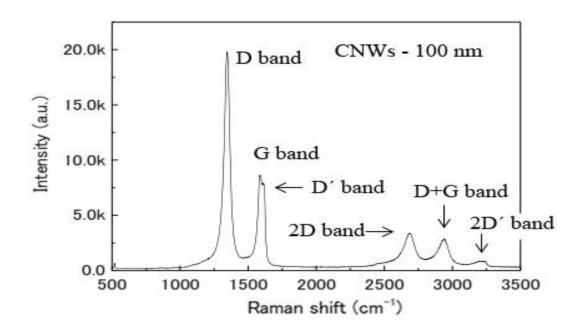


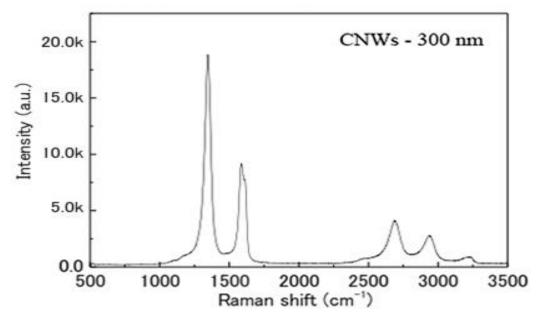
#### Differences between CNW

 $I_D/I_{G_1}I_{D_2}/I_{G_3}$ ,  $I_{2D}/I_{G_3}$  ratios and the position of peaks of Raman spectra for both tested CNWs.

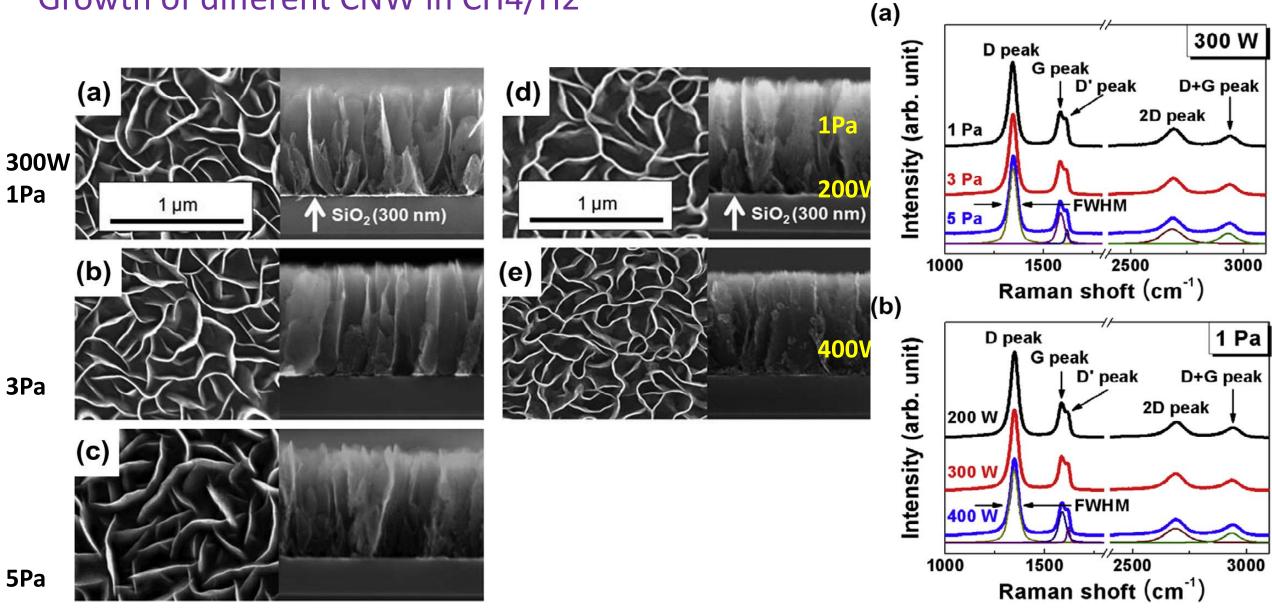
Sampl	$I_D$	$I_{\underline{D'}}$	$I_{2D}$	l <sub>D</sub>	l <sub>G</sub>	l <sub>D</sub> ′	l <sub>2D</sub>
е	$\overline{I_G}$	$I_G$	$I_G$	[cm <sup>-1</sup> ]	[cm <sup>-1</sup> ]	[cm <sup>-1</sup> ]	[cm <sup>-1</sup> ]
CNWs	2.29	0.91	0.40	1357	1597	1624	2693
- 100							
nm							
CNWs	2.04	0.84	0.45	1334	1581	1607	2684
-							
300							
nm							

→ No obvious difference in material structure.

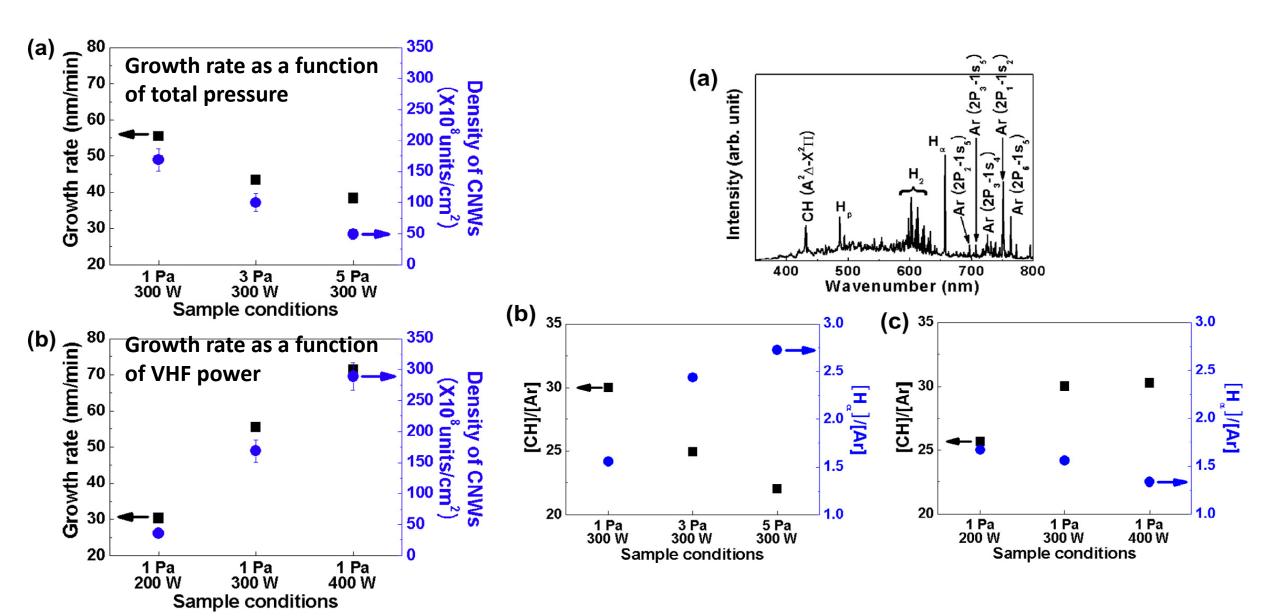




#### Growth of different CNW in CH4/H2



#### Growth of CNW in CH4/H2 and radicals



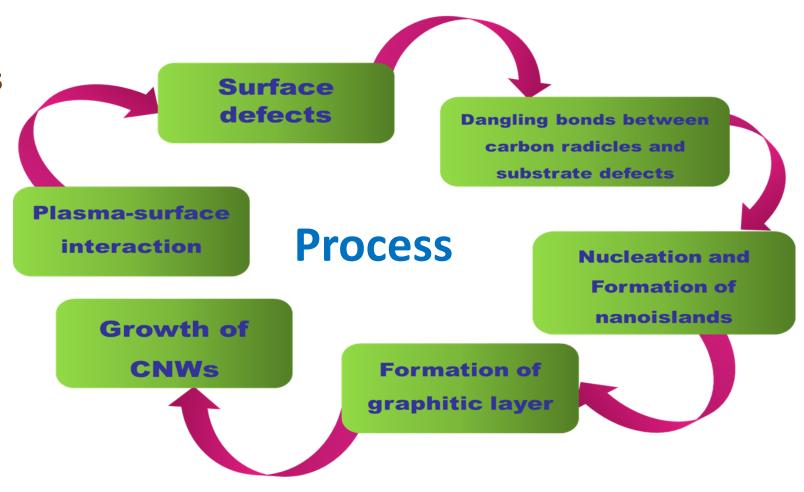
#### Where do we stand?

- 1) The **CNWs density is controlled by the total pressure**, and very-high-frequency CHF power as well as surface temperature.
- 2) No change in the crystal quality (Raman) and the chemical bonding states (XPS) is observed between CNWs
- 3) H-atom density was linearly correlated with the wall density.
- 4) RI-PECVD of CNWs the relative composition in CH and H atom is at least indicator of film quality control.
- 5) We think that **F atoms** from C2F6/H2 plasma might **act as a dopant** at the graphene edges. Consequently, electrical properties could be controlled while maintaining the crystal quality and chemical bonding state.

#### Role of substrate and effect of catalyst

#### **Substrate materials**

- Transition metal substrates
- Dielectric substrates
- Cylindrical substrates



#### **Conclusions**

Growth and processing of vertically aligned 2D carbon nanowalls, and studying the characteristic □ Vertical features for future applications growth **☐** Surface **PECVD** morphology **Finding** Hydrocarbon/ ☐ Enhance Fluorocarbon sources optimum electronic discharge **Functionalization** properties parameters and ☐ Band gap tuning **Growth of** Plasma treatment Finding **CNWS** on carbon applications hydrogels